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ALD type process according to the method vapor-phase pulses of a metal source material, a reducing agent capable of reducing metal source material, and a nitrogen source material capable of reacting with the reduced metal source material are alternately and sequentially fed into a reaction space and contacted with the substrate. According to the invention as the reducing agent is used a boron compound which is capable of forming gaseous reaction byproducts when reacting with the metal source material.

METHOD OF DEPOSITING TRANSITION METAL NITRIDE THIN FILMS

Background of the Invention

5 Field of the Invention

The present invention relates to metal nitride thin films. In particular, the invention concerns a method of growing tungsten nitride thin films by Atomic Layer Deposition (referred to as ALD hereinafter).

10

Description of Related Art

The integration level of components in integrated circuits is increasing, which rapidly brings about a need for a decrease of the size of components and interconnects. Design rules are setting the feature sizes to $\leq 0.2 \mu\text{m}$. Complete film coverage on deep bottoms and
15 vias is hard to obtain.

Integrated circuits contain interconnects which are usually made of aluminium or copper. Especially copper is prone to diffusion to the surrounding materials. Diffusion affects the
20 electrical properties of the circuits and active components may malfunction. The diffusion of metals from interconnects into active parts of the device is prevented with an electrically conductive diffusion barrier layer. Favored diffusion barriers are, e.g., amorphous transition metal nitrides, such as TiN, TaN and WN. The nitrides can be non-stoichiometric because nitrogen is located in interstitial position of the lattice.

25

In the Chemical Vapor Deposition method (referred to CVD hereinafter), the source materials are typically fed to reaction space together, and they react with each other when brought into contact with the substrate. It is also possible to feed one source material containing all desired reactant species to a CVD reactor, and heat it almost to a point where
30 it decomposes thermally. When the heated gas contacts the substrate surface, a cracking

of the film.

Atomic Layer Deposition (ALD) and, originally, Atomic Layer Epitaxy (ALE) is an advanced variation of CVD. The method name was changed from ALE into ALD to avoid possible confusion when discussing about polycrystalline and amorphous thin films. The ALD method is based on sequential self-saturated surface reactions. The method is described in detail in US Patents Nos. 4,058,430 and 5,711,811. The reactor design benefits from the usage of inert carrier and purging gases which makes the system fast.

The separation of source chemicals from each other by inert gases prevents gas-phase reactions between gaseous reactants and enables self-saturated surface reactions leading to film growth which requires neither strict temperature control of the substrates nor precise dosage control of source chemicals. Surplus chemicals and reaction byproducts are always removed from the reaction chamber before the next reactive chemical pulse is introduced into the chamber. Undesired gaseous molecules are effectively expelled from the reaction chamber by keeping the gas flow speeds high with the help of an inert purging gas. The purging gas pushes the extra molecules towards the vacuum pump used for maintaining a suitable pressure in the reaction chamber. ALD provides an excellent and automatic self-control for the film growth.

ALD has recently been used for depositing single layers of titanium nitride TiN (H. Jeon, J.W. Lee, J. H. Koo, Y. S. Kim, Y. D. Kim, D. S. Kim, "A study on the Characteristics of TiN Thin Film Deposited by Atomic Layer Chemical Vapor Deposition method", AVS 46th International Symposium, abstract TF-MoP17, <http://www.vacuum.org/symposium/seattle/technical.html>, to be presented October 27, 1999 in Seattle, USA).

According to Hiltunen et al. NbN, TaN, Ta₃N₅, MoN and Mo₂N can be grown by ALD using metal halogenides as source chemicals (L. Hiltunen, M. Leskelä, M. Mäkelä, L. Niinistö, E. Nykänen, P. Soininen, "Nitrides of Titanium, Niobium, Tantalum and Molybdenum Grown as Thin Films by the Atomic Layer Epitaxy Method", Thin Solid Films, 166 (1988) 149 – 154). The use of additional zinc vapour during the deposition has

J. W. Klaus has disclosed a process for growing tungsten nitride films using an ALD method (J.W. Klaus, "Atomic Layer Deposition of Tungsten and Tungsten Nitride Using Sequential Surface Reactions", AVS 46th International Symposium, abstract TF-TuM6, <http://www.vacuum.org/symposium/seattle/technical.html>, to be presented October 26, 1999 in Seattle, USA). In the process of the publication, tungsten nitride W_2N is grown from WF_6 and NH_3 .

In the art, tungsten compounds have been reduced by using hydrogen (H_2) (US Patent No. 5,342,652 and EP-A2-899 779), silanes, such as SiH_4 (US Patent No. 5,691,235) and chlorosilanes, such as $SiHCl_3$ (US Patent No. 5,723,384).

There are, however, drawbacks related to these prior art methods. Silanes may also react with WF_6 , thus forming tungsten silicides, such as WSi_2 . Hydrogen can reduce a tungsten compound into tungsten metal which has too low vapor pressure for being transported in gas phase onto substrates. Traditional CVD processes may leave significant amounts of impurities in thin films, especially at low deposition temperatures.

Summary of the Invention

It is an object of the present invention to eliminate the problems of the prior art and to provide a novel method of depositing transition metal nitride thin films by an ALD type process. It is a further object of the invention to provide a process for preparing diffusion barriers on metal surfaces in integrated circuits.

The invention is based on the surprising finding that by feeding into a reactor chamber, which contains a substrate, a suitable transition metal compound and, a reducing boron compound pulse and a nitrogen compound, a metal nitride film with low resistivity can be grown. According to the present invention, the reaction between the gaseous boron compound and the metal species reduces the metal compound and gives rise to gaseous reaction byproducts, which easily can be removed from the reaction space.

The method of the present invention for depositing transition metal nitride thin films, comprises the steps of: providing a reactor chamber, which contains a substrate, alternate pulses of a suitable transition metal compound, a reducing boron compound pulse and a nitrogen compound, said boron

compound and said nitrogen compound being fed after the metal compound. Thus, a metal nitride film with low resistivity can be grown in accordance with the principles of ALD method. According to the present invention, the reaction between the gaseous boron compound and the metal species bound to the surface reduces the metal compound and gives rise to gaseous reaction byproducts, which easily can be removed from the reaction space.

A diffusion barrier can be grown in an integrated circuit by depositing, during the manufacture of the integrated circuit, a metal nitride thin film on a dielectric surface or a metal surface present on the silicon wafer blank.

More specifically, the present method is characterized by what is stated in the characterizing part of claim 1.

The process for preparing diffusion barriers is characterized by what is stated in the characterizing part of claim 20.

A number of considerable advantages are achieved with the aid of the present invention. Metal nitride thin films, in particular tungsten nitride thin films, can be grown at low temperatures. The boron compounds used as source materials are easy to handle and vaporise.

As mentioned above, the boron compounds formed as byproducts of the reaction between the metal species and the reducing boron compound are essentially gaseous and they exit the reactor easily when purging with an inert gas. The boron residues in the film are on a very low level, typically below 5 wt-%, preferably 1 wt-% or less and in particular 0.5 wt-% or less. The resistivity of the film is low. The growing rate of the film is acceptable. Also the reaction times are short, and in all it can be said that films can be grown very effectively by means of the present process.

The present method also provides excellent film properties. Thus, the films are grown with excellent conformity even on uneven surfaces and on trenches and vias. The method also provides an excellent and automatic self-control for the film growth.

The metal nitride thin films grown by the present invention can be used, for example, as ion diffusion barrier layers in integrated circuits. Tungsten nitride stops effectively oxygen and increases the stability of metal oxide capacitors. Transition metal nitrides and especially tungsten nitride is also suitable as an adhesion layer for a metal, as a thin film resistor, for stopping the migration of tin through via holes and improving the high-temperature processing of integrated circuits.

Next, the invention is described in detail with the aid of the following detailed description and by reference to the attached drawing.

Brief Description of the Drawing

Figure 1 presents a block diagram of a pulsing sequence according to a preferred embodiment of the invention.

Detailed Description of the Invention

Definitions

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For the purposes of the present invention, a "chemical gaseous deposition process" designates a deposition process in which the reactants are fed to a reaction space in vapor phase. Examples of such processes include CVD and ALD.

25 For the purposes of the present invention, an "ALD type process" designates a process in which deposition of vaporized material onto a surface is based on sequential self-saturating surface reactions. The principle of ALD process is disclosed, e.g., in US 4 058 430.

"Reaction space" is used to designate a reactor or reaction chamber in which the conditions can be adjusted so that the deposition by ALD is possible.

are transported as separate ions, atoms or molecules via vacuum, gaseous phase or liquid phase from the source to the substrate. The thickness of the film depends on the application

and it varies in a wide range, e.g., from one molecular layer to 800 nm, even up to 1000 nm.

The deposition process

5

According to the present invention, metal nitride thin films are prepared by ALD type process.

According to CVD process, a film is grown on a substrate placed in a reaction chamber at
10 elevated temperatures. The principles of CVD are well known to those skilled in the art. The metal source material, the nitrogen source material and the reducing boron compound are typically fed to the reaction space essentially simultaneously, although the duration of the pulsing of the different species may vary. It is also possible to feed a source material comprising both the nitrogen and metal to the reaction space together with the reducing
15 boron compound.

According to the present invention, metal nitride thin films are prepared by the ALD process. Thus, a substrate placed in a reaction chamber is subjected to sequential, alternately repeated surface reactions of at least two vapor-phase reactants for the purpose
20 of growing a thin film thereon. Metal compounds used as source materials are reduced by boron compounds on a substrate maintained at an elevated temperature. The boron compounds, on the other hand, are not incorporated into the film. The reduced metal species react on the surface with gaseous or volatile nitrogen source material.

25 The conditions in the reaction space are adjusted so that no gas-phase reactions, i.e., reactions between gaseous reactants, occur, only surface reactions, i.e., reactions between species adsorbed on the surface of the substrate and a gaseous reactant. Thus, the molecules of the reducing boron compound react with the deposited metal source compound layer on the surface, and the nitrogen source material reacts with the reduced metal compound on
30 the surface.

the reducing agent are alternately and sequentially fed into the reaction space and contacted with the surface of the substrate fitted into the reaction space. The "surface" of the

substrate comprises initially the surface of the actual substrate material which optionally has been pretreated in advance, e.g., by contacting it with a chemical for modifying the surface properties thereof. During the growing of the thin films, the previous metal nitride layer forms the surface for the following metal nitride layer. The reagents are preferably
5 fed into the reactor with the aid of an inert carrier gas, such as nitrogen.

Preferably, and to make the process faster, the metal source material pulse, the reducing boron compound pulse and the nitrogen source material pulse are separated from each other by an inert gas pulse, also referred to as gas purge in order to purge the reaction space
10 from the unreacted residues of the previous chemical. The inert gas purge typically comprises an inactive gas, such as nitrogen, or a noble gas, such as argon.

Thus, one pulsing sequence (also referred to as a "cycle") preferably consists essentially of

- 15 – feeding a vapor-phase pulse of a metal source chemical with the help of an inert carrier gas into the reaction space;
- purging the reaction space with an inert gas;
- feeding a vapor-phase pulse of a boron source chemical with the help of an inert carrier gas into the reaction space;
- purging the reaction space with an inert gas;
- 20 – feeding a vapor-phase pulse of a nitrogen source material into the reaction space; and
- purging the reaction space with an inert gas;

The purging time is selected to be long enough to prevent gas phase reactions and to
25 prevent transition metal nitride thin film growth rates higher than one lattice constant of said nitride per cycle.

The deposition can be carried out at normal pressure, but it is preferred to operate the method at reduced pressure. The pressure in the reactor is typically 0.01 – 20 mbar, preferably 0.1 – 5 mbar. The substrate temperature has to be low enough to keep the bonds
30

between the atoms of the substrate material and the thermal decomposition of the gaseous source materials in gaseous phase. The condensation of the gaseous reactants must be avoided. Further, the temperature must be sufficiently high to provide the activation energy

for the surface reaction. Depending on the reactants and the pressure the temperature of the substrate is typically 200 – 700 °C, preferably 250 – 500 °C.

At these conditions, the amount of reactants bound to the surface will be determined by the surface. This phenomenon is called "self-saturation".

Maximum coverage on the substrate surface is obtained when a single layer of metal source chemical molecules is adsorbed. The pulsing sequence is repeated until a metal nitride film of predetermined thickness is grown.

The source temperature is preferably set below the substrate temperature. This is based on the fact that if the partial pressure of the source chemical vapor exceeds the condensation limit at the substrate temperature, controlled layer-by-layer growth of the film is lost.

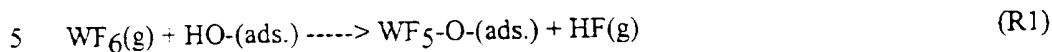
The amount of time available for the self-saturated reactions is limited mostly by the economical factors such as required throughput of the product from the reactor. Very thin films are made by relatively few pulsing cycles and in some cases this enables an increase of the source chemical pulse times and, thus, utilization of the source chemicals with a lower vapor pressure than normally.

The substrate can be of various types. Examples include silicon, silica, coated silicon, copper metal, and various nitrides, such as metal nitrides. Conventionally, the preceding thin film layer deposited will form the substrate surface for the next thin film.

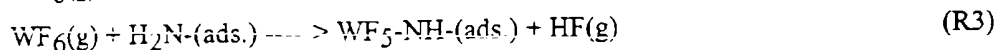
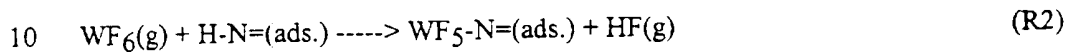
The present method provides for growing of conformal layers in geometrically challenging applications. As mentioned above, it is possible to produce diffusion barriers on dielectric (e.g. silica or nitride) or metal (e.g. copper) surfaces in integrated surfaces. In these cases, said surfaces form the substrates for the growing of the metal nitride thin films.

The metal source material can attach on a nitride surface more easily if there are certain conditions. Table 1 of the present document presents suggested reaction equations of

Silicon wafers have a native oxide on top. The silica (SiO_2) layer may be just a few molecular layers thick. On a silica surface there are "-OH" groups which can serve as reactive surface sites.

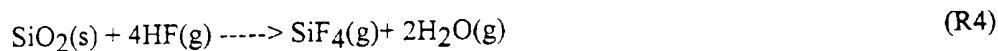


During the growth process, the metal source compound attaches to the nitride surface. Suggested reaction equations for WF_6 are presented in R2 and R3.



It is of importance that the process parameters are carefully optimised to protect silicon wafer against corrosion especially during the first phase of the nitride growth, since the evolved HF gas can attack silica and form volatile silicon tetrafluoride.

15



The uncovered silicon is prone to further, undesired, reactions.

20

The metal source materials most typically used are volatile or gaseous compounds of transition metals, i.e., elements of groups 3, 4, 5, 6, 7, 8, 9, 10, 11 and/or 12 (according to the system recommended by IUPAC) in the periodic table of elements. In particular, the film consists essentially of W, Ti, Zr, Hf, V, Nb, Ta, Cr and/or Mo nitride(s) and thus gaseous or volatile compounds of these are preferably used in the method of the present invention.

25

Since the properties of each metal compound vary, the suitability of each metal compound for the use in the process of the present invention has to be considered. The properties of the compounds are found, e.g., in N. N. Greenwood and A. Earnshaw, Chemistry of the

30

Group 5-10 Elements, Pergamon Press, 1986

The metal source material (as well as the technique of the compound and the metal source material) has to be chosen so that the requirements for sufficient vapor pressure, the

above-discussed criteria of sufficient thermal stability at substrate temperature and sufficient reactivity of the compounds are fulfilled.

Sufficient vapor pressure means that there must be enough source chemical molecules in the gas phase near the substrate surface to enable fast enough self-saturated reactions at the surface.

In practice, sufficient thermal stability means that the source chemical itself must not form growth-disturbing condensable phases on the substrates or leave harmful levels of impurities on the substrate surface through thermal decomposition. Thus, one aim is to avoid non-controlled condensation of molecules on substrates.

Further selecting criteria include the availability of the chemical at high purity, and the ease of handling, inter al., without severe precautions.

15

Typically, suitable metal source materials can be found among halides, preferably fluorides, chlorides, bromides or iodides, or metal organic compounds, preferably alkylaminos, cyclopentadienyls, dithiocarbamates or betadiketonates of desired metal(s).

20 According to a preferred embodiment of the invention, tungsten nitride (W_xN_y , referred to as WN hereinafter) is grown. Then, tungsten source chemical is a tungsten compound selected according to the above criteria. Preferably, the tungsten source material is selected from the group comprising

- a halide such as WF_x , WCl_y , WBr_m or WI_n wherein x, y, m and n are integers from 1 to 6, in particular WF_6 ;
- a carbonyl such as tungsten hexacarbonyl $W(CO)_6$ or tricarbonyl(mesitylene)tungsten;
- cyclopentadienyl such as bis(cyclopentadienyl)tungsten dihydride, bis(cyclopentadienyl)tungsten dichloride or bis(cyclopentadienyl)ditungsten hexacarbonyl; and

30

According to a preferred embodiment transition metal nitrides are mixed so that in the growing process two or more different metal source materials are used. For example, tungsten nitride can be mixed with TiN.

- 5 The metal reactant will react with the substrate surface forming a (covalent) bond to the surface bonding groups. The adsorbed metal species will contain a residue of the reactant compound, such as halogen or hydrocarbon. According to the present invention, this residue reacts with the gaseous boron compound, which reduces the metal compound on the surface.

10

- The reducing strengths of the boron compounds vary. Thus, some boron compounds may reduce the metal compound to elemental metal, and others to a certain oxidation state. It is important that only those metals which are reactive with the nitrogen compounds also in their elemental form are reduced to metals. Typically, the oxidation state of the metal source compound is reduced so that the metal on the surface is in a form of a compound. 15 The metal compounds react with the nitrogen source materials easily forming metal nitrides.

- The boron sources are selected bearing in mind the same criteria as for the metal source materials. In general, the boron compound can be any volatile, thermally sufficiently stable 20 and reactive boron compound capable of reducing the metal species bonded to the surface.

- The reactions of different metal source materials with one and same reducing agent (and vice versa) can lead to different reaction (by)products. According to the present invention, 25 the metal source material and boron compound are selected so that the resulting boron compound(s) is (are) gaseous. By this is meant that the compound formed is gaseous enough to be moved from the reaction space with the aid of the inert purging gas, and, on the other hand, does not decompose, e.g., catalytically or thermally, to condensable species. In all, byproducts will not remain as impurities in the films. If a reactive site on the surface 30 is contaminated, the growing rate of the film decreases. By selecting the metal source material and boron compound suitably, the growing rate of the film does not decrease. In particular, it is preferred that the reaction of the metal source material and boron compound does not lead to a solid byproduct. An example of an unsuitable pair is TiCl_4 and triethyl boron, the reaction thereof not leading to desired results.

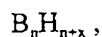
The selection can be facilitated with computer programs having a sufficiently extensive thermodynamics database, which enables to check the reaction equilibrium and thus predict which reactants have thermodynamically favorable reactions. An example of this kind of programs is HSC Chemistry, version 3.02 (1997) by Outokumpu Research Oy, Pori, Finland.

A vast range of boron chemicals makes it possible to choose suitable reducing strength and avoid boride formation. It is possible to use one or more boron compounds in the growing of one and same thin film.

Preferably, one or more of the following boron compounds is used:

Boranes having formula (I)

15



wherein n is an integer from 1 to 10, preferably from 2 to 6, and
 x is an even integer, preferably 4, 6 or 8,

20

or formula (II)



(II)

wherein n is an integer from 1 to 10, preferably from 2 to 6, and
 m is an integer different than n , m being from 1 to 10, preferably from 2 to 6.

25

Boranes according to formula (I) are exemplified by *nido*-boranes (B_nH_{n+4}), *arachno*-boranes (B_nH_{n+6}) and *hypho*-boranes (B_nH_{n+8}). Of the boranes according to formula (II), examples include *conjuncto*-boranes (B_nH_m). Also borane complexes, such as $(CH_3CH_2)_3N \cdot BH_3$ can be used.

30

complexes

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Boron halides with high boron/halide ratio, such as B_2F_4 , B_2Cl_4 and B_2Br_4 .

Halogenoboranes according to formula (III)

5 B_nX_n , (III)

wherein X is Cl or Br and

$n = 4, 8-12$ when $X = Cl$, and

$n = 7-10$ when $X = Br$

10 Carboranes according to formula (IV)

$C_2B_nH_{n+x}$, (IV)

wherein n is an integer from 1 to 10, preferably from 2 to 6, and

x is an even integer, preferably 2, 4 or 6.

15 Examples of carboranes according to formula (IV) include *closo*-carboranes ($C_2B_nH_{n+2}$), *nido*-carboranes ($C_2B_nH_{n+4}$), and *arachno*-carboranes ($C_2B_nH_{n+6}$).

Amine-borane adducts according to formula (V)

20 R_3NBX_3 , (V)

wherein R is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl or H, and

X is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl, H or halogen,

25 Aminoboranes where one or more of the substituents on B is an amino group according to formula (VI)

R_2N , (VI)

wherein R is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl or substituted or unsubstituted aryl group.

30 An example of suitable aminoborane is $(CH_3)_2NB(CH_3)_2$.

Alkyl borons or alkyl boranes, wherein the alkyl is typically linear or branched $C_1 - C_{10}$ alkyl, preferably $C_2 - C_4$ alkyls. Particularly preferred is triethyl boron $(CH_3CH_2)_3B$, because it is easily vaporized.

- 5 Particularly preferred boron compound is triethyl boron $(CH_3CH_2)_3B$.

The reduced metal species bound on the substrate surface will then be subjected to reaction with a nitrogen-containing compound. The nitrogen compound used as the nitrogen source material is volatile or gaseous and chosen according to the above criteria, including the
10 criterion relating to the reaction byproducts.

Preferably, the nitrogen compound is selected from the group comprising

- ammonia (NH_3) and its salts, preferably halide salt, in particular ammonium fluoride or ammonium chloride;
- 15 - hydrogen azide (HN_3) and the alkyl derivatives of the said compound such as CH_3N_3 ;
- hydrazine (N_2H_4) and salts of hydrazine such as hydrazine hydrochloride;
- alkyl derivatives of hydrazine such as dimethyl hydrazine;
- nitrogen fluoride NF_3 ;
- 20 - hydroxyl amine (NH_2OH) and its salts such as hydroxylamine hydrochloride;
- primary, secondary and tertiary amines such as methylamine, diethylamine and triethylamine; and
- nitrogen radicals such as NH_2^* , NH^{**} and N^{***} , wherein * means a free electron capable of bonding, and excited state of nitrogen (N_2^*).

25

When no reducing agent is used, the nitride film resulting from the above-described process has a N/W molar ratio of greater than 1, i.e., the nitride is mostly in the form WN_2 .

When operating without a reducing agent, it is also possible to feed the nitrogen source material pulse into the reaction space first and the transition metal source material second,

- 30 i.e., use a reverse order of source material pulses. The deposition process ends also in this

case. The structure of the film is different from

that produced according to the process employing no reducing agent, has rather high resistivity.

The following non-limiting examples illustrate the invention.

EXAMPLE 1

5 Tungsten hexafluoride (WF_6) and ammonia (NH_3) were used as source chemicals. Both chemicals are liquefied gases at room temperature and poses high enough vapor pressure without additional heating for the ALD process. Source tubing and the reactor were purged with nitrogen gas which had a purity of 99.9999% (i.e. 6.0). The N_2 gas was prepared from liquid nitrogen. A 200-mm silicon wafer was loaded to an ALD reactor as described in
10 Finnish Patent No. 100409 of assignee. Source chemicals were pulses alternately to the substrates at the reaction chamber. The deposition was started and ended with an NH_3 pulse. The pulsing cycle consisted of the following steps:

NH_3 vapor pulse 0.5 s

15 N_2 gas purge 1.0 s

WF_6 vapor pulse 0.25 s

N_2 purge 0.8 s

The pulsing cycle was repeated for 500 times which produced a 30 nm film with the typical
20 growth rate of 0.6 Å/cycle. The composition, impurities and the thickness of the resulting thin film were analyzed by Electron Diffraction Spectroscopy (referred to as EDS hereinafter). EDS showed an N/W ratio of 1.3 which means that the phase of tungsten nitride was between WN and WN_2 , i.e. rich in nitrogen. Decreasing the growth temperature from 400 °C to 360 °C increased the fluorine content from 2 wt.-% to 4 wt.-%.

25

The resistivity of the tungsten nitride film was obtained by combining the thickness value with the four-point probe measurements. The resistivity of the film grown at 400 °C was 1900 $\mu\Omega\text{cm}$. High resistivity was possibly caused by the high nitrogen content of the film.

30 EXAMPLE 2

as source chemicals. All the chemicals are liquids or liquefied gases at room temperature and poses high enough source vapor pressure without additional heating for the ALD

process. Source tubing and the reactor were purged with nitrogen gas which had a purity of 99.9999% (i.e. 6.0). The N₂ gas was prepared from liquid nitrogen. A 200-mm silicon wafer was loaded to an F200 ALD reactor. Source chemicals were pulses alternately to the substrates at the reaction chamber. The pulsing cycle consisted of the following steps:

5

WF₆ vapor pulse 0.25 s

N₂ purge 0.8 s

(CH₃CH₂)₃B vapor pulse 0.01 s

N₂ gas purge 0.5 s

10

NH₃ vapor pulse 0.25 s

N₂ gas purge 0.5 s

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The pulsing cycle was repeated for 500 times resulting in a 30-nm film at 360 °C. The samples were analyzed by EDS for thickness and composition. The thin film consisted of tungsten and nitrogen while boron could not be seen in detectable amounts. There was 3 wt.-% of fluorine as an impurity in the film. The resistivity of the tungsten nitride film was obtained by combining the thickness value with the four-point probe measurements. The resistivities were 130 - 160 μΩcm.

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The inventors assume that the boron chemical acted as a reducing agent and removed fluorine from tungsten fluoride. The benefit of this boron chemical is that possible byproducts such as BF₃ and CH₃CH₂F are gaseous at the deposition temperature and do not disturb the nitride growth.

Claims:

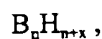
1. A method of growing metal nitride thin film on a substrate by an Atomic Layer Deposition (ALD) type process, wherein vapour-phase pulses of
 - 5 – a metal source material,
 - a reducing agent capable of reducing the metal source material, and
 - a nitrogen source material capable of reacting with the reduced metal source material,are alternately and sequentially fed into a reaction space and contacted with the substrate,
 - 10 characterized by
 - using as the reducing agent a boron compound which is capable of forming gaseous reaction by-products when reacting with the metal source material.
2. The method according to claim 1, wherein the reducing agent is capable of forming
 - 15 gaseous reaction by-products when reacting with the metal source material bound to the surface.
3. The method according to claim 1 or 2, wherein the pulsing cycle consists essentially of the steps of
 - 20 – feeding a vapor-phase pulse of metal source chemical with the help of an inert carrier gas into the reaction space;
 - purging the reaction space with an inert gas;
 - feeding a vapor-phase pulse of a boron source chemical with the help of an inert carrier gas into the reaction space;
 - 25 – purging the reaction space with an inert gas;
 - feeding a vapor-phase pulse of the nitrogen source chemical into the reaction space; and
 - purging the reaction space with an inert gas.
- 30 4. The method according to any of claims 1 - 3, wherein the metal in the metal source is selected from the group consisting of W, Mo, Cr, Ta, Nb, V, Hf, Zr and Ti.
5. The method according to claim 1, wherein the metal compound in the metal source

material is selected from the group comprising halides, preferably fluorides, chlorides, bromides or iodides, and metal organic compounds, preferably alkylaminos, cyclopentadienyls, dithiocarbamates or betadiketonates.

- 5 6. The method according to claim 4 or 5, wherein the metal source material is a tungsten compound selected from the group comprising
- tungsten halides, preferably WF_x , WCl_y , WBr_m or WI_n wherein x, y, m and n are integers from 1 to 6, in particular WF_6 ;
 - tungsten carbonyls, preferably tungsten hexacarbonyl $W(CO)_6$ or
 - 10 tricarbonyl(mesitylene)tungsten;
 - tungsten cyclopentadienyls such as bis(cyclopentadienyl)tungsten dihydride, bis(cyclopentadienyl)tungsten dichloride or bis(cyclopentadienyl)ditungsten hexacarbonyl; and
 - tungsten β -diketonate.

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7. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising boranes of formula (I)



- 20 wherein n is an integer from 1 to 10, preferably from 2 to 6, and
x is an even integer, preferably 4, 6 or 8,

and formula (II)



- wherein n is an integer from 1 to 10, preferably from 2 to 6, and
25 m is an integer different than n, m being from 1 to 10, preferably from 2 to 6,
and complexes thereof.

8. The method according to claim 7, wherein the boranes are selected from the group comprising *nido*-boranes of formula B_nH_{n+4} , *arachno*-boranes of formula B_nH_{n+6} , *hypho*-
30 boranes of formula B_nH_{n+8} and *conjuncto*-boranes B_nH_m , wherein n and m are the same as in claim 6.

from the group comprising carboranes according to formula (IV)



wherein n is an integer from 1 to 10, preferably from 2 to 6, and
 x is an even integer, preferably 2, 4 or 6.

- 5 10. The method according to claim 9, wherein the carboranes are selected from the group comprising *closo*-carboranes ($C_2B_nH_{n+2}$), *nido*-carboranes ($C_2B_nH_{n+4}$) and *arachno*-carboranes ($C_2B_nH_{n+6}$), wherein n is the same as in claim 8.

11. The method according to any of claims 1 - 6, wherein the boron compound is selected
 10 from the group comprising amine-borane adducts according to formula (V)



wherein R is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl or H, and
 X is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl, H or halogen.

- 15 12. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising aminoboranes where one or more of the substituents on B is an amino group according formula (VI)



- wherein R is linear or branched $C_1 - C_{10}$, preferably $C_1 - C_4$ alkyl or substituted or
 20 unsubstituted aryl group.

13. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising alkyl borons and alkyl boranes, wherein the alkyl is linear or branched $C_1 - C_{10}$ alkyl, preferably $C_2 - C_4$ alkyl.

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14. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising boron halides having a high boron/halide ratio, preferably B_2F_4 , B_2Cl_4 and B_2Br_4 .

- 30 15. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising halogenoboranes of formula (III)

wherein x is 4 or 6, and

$n = 4, 8-12$ when $X = Cl$, and

$n = 7-10$ when $X = \text{Br}$.

16. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising cyclic borazine ($-\text{BH}-\text{NH}-$)₃ and the volatile derivatives thereof.
17. The method according to any of claims 1 - 6, wherein the boron compound is selected from the group comprising borane halides and complexes thereof.
18. The method according to any of the preceding claims, wherein the nitrogen source material is selected from the group comprising
 - ammonia (NH_3) and its salts, preferably halide salt, in particular ammonium fluoride or ammonium chloride;
 - hydrogen azide (HN_3) and the alkyl derivatives thereof, preferably CH_3N_3 ;
 - hydrazine (N_2H_4) and salts of hydrazine such as hydrazine hydrochloride;
 - alkyl derivatives of hydrazine, preferably dimethyl hydrazine;
 - nitrogen fluoride NF_3 ;
 - hydroxyl amine (NH_2OH) and salts thereof, preferably hydroxylamine hydrochloride;
 - primary, secondary and tertiary amines, preferably methylamine, diethylamine and triethylamine; and
 - nitrogen radicals such as NH_2^* , NH^{**} , N^{***} , wherein $*$ is a free electron capable of bonding, and excited state of nitrogen (N_2^*).
19. The method according to any of the preceding claims, wherein the substrate is selected from the group comprising silicon, silica, coated silicon, copper metal, and nitrides.
20. Process for preparing a diffusion barrier in an integrated circuit, comprising depositing during the manufacture of the integrated circuit a metal nitride thin film on a dielectric surface or a metal surface of the integrated circuit by a method in accordance with any of the claims 1 - 19.

growing transition metal nitride thin films by an ALD process.

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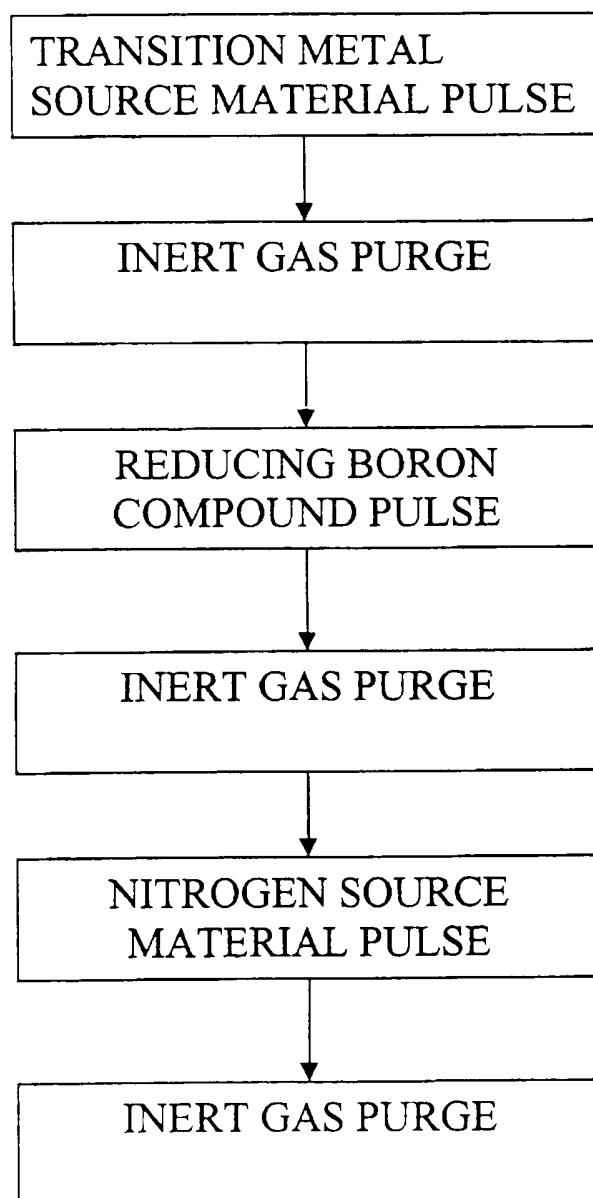


Fig. 1

INTERNATIONAL SEARCH REPORT

International application No.

PCT/FI 00/00895

A. CLASSIFICATION OF SUBJECT MATTER

IPC7: C23C 16/44, C23C 16/00, C30B 25/02, H05K 3/00

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

IPC7: C23C, C30B, H01L, H05K

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

SE,DK,FI,NO classes as above

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	J. ELECTROCHEMICAL SOCIETY, Volume 142, No 8, August 1995, Mikko Ritala et al, "Atomic Layer Epitaxy Growth of TiN Thin Films" page 2731 - page 2737 --	1-21
A	APPLIED SURFACE SCIENCE, Volume 82/83, 1994, Kai-Erik Elers et al, "NbC15 as a precursor in atomic layer epitaxy", page 468 - page 474, see especially page 469 - 471 and 473 --	1-21

☒ Further documents are listed in the continuation of Box C.☐ See patent family annex.

* Special categories of cited documents:

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"X" document of particular relevance: the claimed invention cannot be considered novel, or cannot be considered to involve an inventive step when the document is taken alone

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16 January 2001

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INTERNATIONAL SEARCH REPORT

International application No.

PCT/FI 00/00895

C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT		
Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	<p>JP 8264530 A (FUJITSU LTD) 1996-10-11 (abstract) World Patents Index (online). London, U.K.: Derwent Publications, Ltd. (retrieved on 2001-01-16). Retrieved from: EPO WPI Database. DW199651, Accession No. 1996-511211; & JP 8264530 (FUJITSU LTD) 1997-02-28 (abstract). (online)(retrieved on 2001-01-16). Retrieved from: EPO PAJ Database</p> <p style="text-align: center;">-- -----</p>	1-21